analysis indicated that the $\mathrm{Fe}^{2+}$ ions have a higher concentration at the $M(1)$ site than at $M(2)$, and that the $\mathrm{Na}^{+}$site is approximately half-vacant, giving the chemical formula $\mathrm{Na}_{0.56} \mathrm{Fe}_{0.28}^{\mathrm{II}} \mathrm{Ti}_{1.72}^{1 \mathrm{~V}} \mathrm{O}_{4}$. Though some of the $\mathrm{Fe}^{2+}$ ions might be oxidized to $\mathrm{Fe}^{3+}$, the amount should not be large, since the final difference Fourier maps are quite flat around the $\mathrm{Na}^{+}$and ( $\mathrm{Ti}^{4+}, \mathrm{Fe}^{2+}$ ) sites.

The ( $\mathrm{Ti}, \mathrm{Fe}$ ) $\mathrm{O}_{6}$ octahedra share edges, constructing double chains parallel to [010]. The chains are further joined laterally with each other by sharing corners of the octahedra to form a three-dimensional framework with ditrigonal tunnels running parallel to [010]. The $M(1)-\mathrm{O}$ distances range from 1.98 to $2.01 \AA$ (mean $2.00 \AA$ ), while the $M(2)-\mathrm{O}$ distances are in the range from 1.95 to $2.04 \AA$ (mean $1.98 \AA$ ). These values are in agreement with the ( $\mathrm{Ti}, \mathrm{Fe}$ )-O distances found in $\mathrm{Na}_{0.90} \mathrm{Fe}_{0.90} \mathrm{Ti}_{1 \cdot 10} \mathrm{O}_{4}$ (Mumme \& Reid, 1968) and freudenbergite (Ishiguro, Tanaka, Marumo, Ismail, Hirano \& Somiya, 1978). In fact, the ionic radii given by Shannon \& Prewitt (1969) suggest that the ( $\mathrm{Ti}, \mathrm{Fe}$ )- O distance in the present crystal should be longer than those in the above two crystals by only 0.005 and $0.013 \AA$ respectively. It is notable that the $\mathrm{Fe}^{2+}$ ion with a larger ionic radius than $\mathrm{Ti}^{4+}$ is more concentrated at the $M(1)$ site which has a slightly larger octahedral environment than $M(2)$.

The $\mathrm{Na}-\mathrm{O}$ distances range from 2.39 to $2.63 \AA$ (mean $2.48 \AA$ ). Since the $\mathrm{Na}^{+}$ions lie in the wide tunnel spaces and approximately one half of their sites are
vacant, this compound is expected to show ionic conduction by $\mathrm{Na}^{+}$ions in the [010] direction.

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# Lead Zirconium Sulphide 

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#### Abstract

PbZrS}_{3}\), Pnma, $a=9.0134$ (7), $b=$ 3.7660 (2), $c=13.9237$ (10) $\AA, Z=4$. Structure refinement based on neutron diffraction powder diagrams established the modified $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ structure of $\mathrm{PbZrS}_{3}$. The compound is isostructural with $\mathrm{PbSnS}_{3}$.


Introduction. We investigated the crystal structure of $\mathrm{PbZrS}_{3}$ as a part of a research programme on the structural relations of compounds $A B X_{3}$, where $A$ represents an alkaline or alkaline-earth metal, $B$ a transition metal and $X$ a halogen or S .
Sterzel \& Horn (1970) reported the structures of $\mathrm{PbZrS}_{3}$ and $\mathrm{PbHfS}_{3}$, but their samples were contami-
nated with PbS . Their resulting X-ray powder pattern could be indexed like $\mathrm{PbSnS}_{3}$ (Yamaoko \& Okai, 1970), Table 1.

Table 1. Crystal axes for $A B \mathrm{~S}_{3}$

|  | $a(\mathrm{~A})$ | $b$ ( ( ) | $c\left(\begin{array}{l}\text { ( }\end{array}\right.$ | Reference |
| :---: | :---: | :---: | :---: | :---: |
| $\mathrm{PbZrS}_{3}$ | 9.031 (6) | 3.770 (4) | 13.919 (9) $\}$ | Sterzel \& Horn (1970), |
| $\mathrm{PbHfS}_{3}$ | 8.989 (2) | $3 \cdot 738$ (2) | 13.924 (4) $\}$ | omitting their PbS pattern |
| PbZrS 3 | 9.037 (5) | 3.77 (2) | 13.926 (5) | Yamaoko (1972) |
| $\mathrm{PbZrS}_{3}$ | 9.0134 (7) | 3.7660 (2) | 13.923 (10) | This work |
| $\mathrm{PbSnS}_{3}$ | 8.738 | 3.792 | 14.052 | Jumas et al. (1972) |
| $\mathrm{Sn}_{2} \mathrm{~S}_{3}$ | $8 \cdot 864$ (6) | 3.747 (1) | 14.020 | Mootz \& Puhl (1972) |

Yamaoko (1972) reported the structure of $\mathrm{PbZrS}_{3}$, prepared under a pressure of 20 kbar at $800-900^{\circ} \mathrm{C}$; the X-ray powder diagram is similar to that of $\mathrm{Sn}_{2} \mathrm{~S}_{3}$ (Mootz \& Puhl, 1967).

We prepared $\mathrm{PbZrS}_{3}$ and $\mathrm{PbHfS}_{3}$ by firing stoichiometric amounts of the binary sulphides in evacuated sealed quartz tubes at $800^{\circ} \mathrm{C}$ for a week. The X-ray diffraction patterns of both compounds could be indexed leading to the space groups Pnma or $P n 2_{1} a$, in agreement with the data of Yamaoko (1972). Neutron diffraction powder data of $\mathrm{PbZrS}_{3}$ were collected on a neutron diffractometer at the HFR reactor in Petten. The neutron wavelength was $2.570 \AA$. Soller slits with a horizontal divergence at $30^{\prime}$ were placed before the monochromator and in front of the $\mathrm{BF}_{3}$ counter.

With neutron diffraction data in the range $5.4<$ $2 \theta<138^{\circ}$, a structure refinement based on the Rietveld (1969) profile method has been carried out in space group Pnma by minimizing the residual function $\chi^{2}=\sum_{i} w_{i}\left|y_{i}(\mathrm{obs})-(1 / c) y_{i}(\mathrm{calc})\right|^{2}$. The $R$ factors are defined by:

$$
\begin{aligned}
R_{\text {nuclear }} & =100 \sum\left|I(\mathrm{obs})-\frac{1}{c} I(\mathrm{calc})\right| / \sum I(\mathrm{obs}), \\
R_{\text {profle }} & =100 \sum\left|y(\mathrm{obs})-\frac{1}{c} y(\mathrm{calc})\right| / \sum y(\mathrm{obs}), \\
R_{\text {weight }} & =100 \sqrt{ }\left[\sum w \mid y(\mathrm{obs})\right. \\
& \left.-\left.\frac{1}{c} y(\mathrm{calc})\right|^{2} / \sum w|y(\mathrm{obs})|^{2}\right] .
\end{aligned}
$$

Table 2. Structural parameters for $\mathrm{PbZrS}_{3}$ in spacegroup Pnma with overall isotropic temperature factor $B$

|  |  | $x$ | $y$ | $z$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Pb | 4(c) | 0.5167 (3) | 0.25 | $0 \cdot 826$ |  |
| Zr | 4(c) | $0 \cdot 1669$ (4) | 0.25 | 0.0498 |  |
| S(1) | 4(c) | 0.2738 (9) | 0.25 | 0.2142 |  |
| S(2) | 4(c) | 0.1637 (11) | 0.25 | 0.4908 |  |
| S(3) | 4(c) | 0.0113 (14) | 0.25 | 0.8932 |  |
| $R_{\text {nuclear }}$ | $\begin{aligned} & 4.55 \\ & B\left(\AA^{2}\right) \end{aligned}$ | $\begin{gathered} R_{\text {profile }} \\ 0.70(5) \end{gathered}$ | 7.95 <br> Residue | $\begin{gathered} R_{\text {welght }} \\ 4.25 \end{gathered}$ | $10 \cdot 08$ |

Table 3. Interatomic distances ( $\AA$ )
Distances marked with an asterisk occur in pairs.

|  |  |  |  |
| :--- | :--- | :--- | :--- |
| $\mathrm{Pb}-\mathrm{S}(1)$ | $2.728(6)^{*}$ | $\mathrm{Zr}-\mathrm{S}(1)$ | $2.481(9)$ |
|  | $3.579(8)^{*}$ | $\mathrm{Zr}-\mathrm{S}(2)$ | $2.558(7)^{*}$ |
| $\mathrm{~Pb}-\mathrm{S}(2)$ | $2.879(9)$ | $\mathrm{Zr}-\mathrm{S}(3)$ | $2.599(8)^{*}$ |
|  | $3.383(8)^{*}$ |  | $2.590(10)$ |
| $\mathrm{Pb}-\mathrm{S}(3)$ | $3.055(9)$ |  |  |

$I(\mathrm{obs}), I($ calc $)=$ observed and calculated integrated intensity of each reflection, $y$ (obs), $y$ (calc) $=$ observed and calculated profile data point, $w=$ statistical weight allotted to each data point, $c=$ scale factor.

The coherent scattering lengths ( $\times 10^{-12} \mathrm{~cm}^{2}$ ) were 0.940 for $\mathrm{Pb}, 0.71$ for Zr and 0.28 for S (Bacon, 1972). The results are shown in Tables 2 and 3. The space group $P n 2_{1} a$ gave no significantly better results.

Discussion. In the $A B S_{3}$ compounds with large radius of $A$ a number of structures are realized; the (distorted) perovskites, the $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ structure and the $\mathrm{TlPbI}_{3}$ ( $\mathrm{NdYbS}_{3}$ ) structure (Stoeger, 1977).

The structure of $\mathrm{PbZrS}_{3}$, and most likely that of $\mathrm{PbHfS}_{3}$, is strongly related to the $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ type (Strukturbericht, type E 24), containing columns of double edge-sharing octahedra linked by Pb ions in tricapped trigonal-prismatic coordination. However, two of the anions, $S(1)$ and $S(3)$, have poor charge compensation in this structure. This may result in smaller $\mathrm{Pb}-\mathrm{S}(1)$ and longer $\mathrm{Pb}-\mathrm{S}(3)$ distances, as is the case in $\mathrm{KCdBr}_{3}$ (Natarajan Iyer, Faggiani \& Brown, 1977). However, in $\mathrm{PbZrS}_{3}$ the Pb is moved to $S(2)$ and one pair of $S(1)$, resulting in an asymmetric position of Pb in the prismatic coordination and a hole for a lone pair. This feature is also found in $\mathrm{PbSnS}_{3}$ (Jumas, Ribes, Philippot \& Maurin, 1972) and $\mathrm{Sn}_{2} \mathrm{~S}_{3}$.

In contrast, the sulphides $A \mathrm{ZrS}_{3}$, with $A=\mathrm{Ca}, \mathrm{Sr}, \mathrm{Eu}$ and Ba (Lelieveld \& IJdo, 1978) are all distorted perovskite-type structures.

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